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European Patent Office

Office européen des brevets



(11) EP 0 984 443 A1

(12)

EUROPEAN PATENT APPLICATION

published in accordance with Art. 158(3) EPC

(43) Date of publication: 08.03.2000 Bulletin 2000/10

(21) Application number: 99906469.4

(22) Date of filing: 24.02.1999

(51) Int. Cl.7: **G11B 7/24**

(86) International application number: PCT/JP99/00850

(87) International publication number:WO 99/44199 (02.09.1999 Gazette 1999/35)

(84) Designated Contracting States: **DE FR GB**

(30) Priority: 24.02.1998 JP 4258798

12.10.1998 JP 28982698 13.02.1999 JP 7424699

(71) Applicant: Sony Corporation Tokyo 141-0001 (JP) (72) Inventors:

Kuroda, Yuji
 Shinagawa-ku, Tokyo 141-0001 (JP)

 Kasami, Yutaka Shinagawa-ku, Tokyo 141-0001 (JP)

(74) Representative:
Müller, Frithjof E., Dipl.-Ing. et al
Patentanwälte
MÜLLER & HOFFMANN,
Innere Wiener Strasse 17
81667 München (DE)

(54) OPTICAL RECORDING MEDIUM

(57) An optical recording medium having a recording layer made of a phase-transition material. The ratio (Ac/Aa) of the absorption factor (Ac) of when the recording layer is in a crystalline phase to the absorption factor (Aa) of when it is in an amorphous phase is 0.9 or over. A crystallization-promoting layer for promoting crystallization of the phase-transition material is formed in contact with at least one of the sides of the recording layer. Control of absorption factor and promotion of crystallization are both achieved, so that the differences of physical properties of crystalline and amorphous phases are properly compensated, and good direct overwrite is realized even under high-speed and high-density condition.

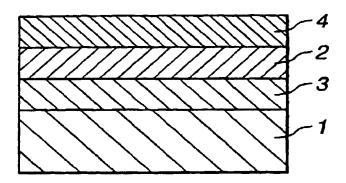


FIG.3

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Description

Technical Field

[0001] The present invention relates to a phase-change type optical recording medium, and more particularly to a phase-change type optical recording medium which permits high-speed direct overwriting.

Background Art

[0002] As a practical example of an optical disc made of a phase-change recording material and permitting overwriting, a so-called DVD-RAM has been marketed. Thus, a linear velocity of 6 m/second, a bit length of 0.41 μm/second, a track pitch of 0.74 μm, a laser wavelength of about 650 nm, a data transfer rate of 11 Mbps and a recording capacity of 2.6 GB have been realized.

[0003] To realize a larger capacity and a higher transfer rate, it is effective to reduce the spot size of a recording laser beam to raise the linear recording velocity. As a specific method of reducing the spot size of the recording laser beam, a method of shortening the wavelength of the laser beam and a method of enlarging the numerical aperture of the object lens are exemplified.

[0004] When both of the method of shortening the wavelength of the laser beam and the method of enlarging the numerical aperture are employed, the spot size can furthermore be reduced as compared with a structure in which only either of the methods is employed. When a violet laser beam having a wavelength of about 400 nm is employed as the light source and an objective lens having a numerical aperture NA of 0.85 is employed, higher density recording is theoretically permitted.

[0005] On the other hand, conditions under which high speed direct overwriting is permitted for the phase-change optical disc become severer because of the following reasons.

[0006] In general, the phase-change optical disc is irradiated with a large-power laser beam so that the temperature of the recording layer is raised to a level not lower than the melting point of the recording layer. Thus, the recording layer is melted, and then the temperature is rapidly lowered. As a result, writing is performed. The recorded marks are crystallized, that is, erased because the recording layer is maintained in a temperature range between a temperature at which the crystallization of the recording layer is started and the melting point for a time required for the recording layer to be formed into crystal.

[0007] When the foregoing method of shortening the wavelength of the laser beam and the method of enlarging the numerical aperture of the objective lens are employed, the temperature of a position on the optical disc is changed in a time shorter than the time required for the conventional structure under a condition that the linear recording velocity is high. FIG. 1 shows a result of calculations of a process of change of the temperature at one point on an optical disc as time elapses. As can be understood from FIG. 1, as the linear velocity is raised, and as the numerical aperture NA of the objective lens is enlarged, time for which the temperature not lower than the crystallizing temperature (for example, assuming that the temperature is 400°C) is maintained is shortened.

[0008] Therefore, a recording material having the crystallizing velocity which is the same as that of the conventional material encounters difficulty in crystallizing amorphous marks, that is, in erasing the amorphous marks.

[0009] So-called direct overwriting (DOW) is considered with which the power level of one laser beam is controlled in terms of time to perform recording. If the spot size is too small or if the linear recording velocity is too high, distortion of the shapes of marks is enhanced owning to the difference in the physical properties between amorphous and crystal which is a peculiar problem for phase-change recording. That is, when a next mark is overwritten on a written recording mark, the size of a recording mark is, under the foregoing condition, enlarged as compared with a case in which the next mark is newly written on a crystal portion in which no data has been written.

[0010] The foregoing fad is caused from the following differences from the crystal phase: response (an optical constant) of the amorphous phase with respect to a laser beam; a process of conduction of heat (heat conductivity) generated owning to reactions with the laser beam; and a method of using generated heat (latent heat is not required when dissolving is performed). When the spot size is large and the linear velocity is low, change in the temperature of the recording film is moderate in terms of time. Heat is conducted prior to movement of the existing mark to the laser beam. Thus, the mark is previously maintained at the crystallizing temperature. Therefore, a state similar to a state in which no mark exists can be realized (previous crystallization). Hence it follows that the foregoing problem does not arise.

[0011] The foregoing problems virtually inhibit a usual phase-change recording disc having a four-layered structure consisting of a $ZnS-SiO_2$ layer, a recording layer, a $ZnS-SiO_2$ layer and a reflecting layer to have higher density and higher transfer rate. For example, deterioration in the value of jitters as the linear velocity is raised as shown in FIG. 2 has been confirmed as a result of experiments.

[0012] As a means for overcoming the foregoing problem, it might be considered feasible to raise the crystallizing velocity of the recording layer. That is, time required to complete crystallization is shortened so as to raise the erasing

ratio and facilitate previous crystallization.

[0013] However, a material, the phase of which is reversibly changed, which has crystallizing velocity higher than those of the conventional materials and which can be applied to an optical disc, cannot virtually be obtained.

[0014] As an alternative to raising the crystallizing velocity of the recording material, techniques for raising the crystallizing velocity of the recording layer by providing a crystallization enhancing material, which is effective to raise the crystallizing velocity of amorphous, such that the material is in contact with the recording layer have been disclosed. The foregoing techniques have been disclosed in Japanese Patent Laid-Open No. 1-92937, Japanese Patent Laid-Open No. 6-195747 and Japanese Patent Application No. 9-532424.

[0015] The foregoing methods, however, encounter problem in that the preservation stability of the recording marks deteriorates in a temperature range under a condition of daily use. Another problem arises in that the recording mark is undesirably erased even with a laser beam for reproducing data.

[0016] As a method structured individually from the foregoing enhancement of crystallization, methods of a type inverting the absorption ratio of amorphous which is usually higher than that of crystal is inverted (the temperature raising velocity of the crystal portion and that of the amorphous portion are balanced) by controlling the lamination structure of the thin optical film have been disclosed. The foregoing method is a so-called absorption-ratio control method which has been disclosed in Japanese Patent Laid-Open No. 8-124218 and Japanese Patent Laid-Open No. 9-91755.

[0017] The foregoing method, however, suffers from a problem in that the degree of freedom of optical design (absorptance, reflectance and so forth) decreases. Another problem arises in that satisfactory durability against repeated overwriting cannot be realized.

[0018] What is worse, conditions in which the diameter of the spot is reduced and the linear velocity is high encounter unsatisfactory basic erasing performance (for example, erasing ratio obtained with DC light) which is required to obtain an effect of controlling the absorptance.

Disclosure of the Invention

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[0019] An object of the present invention is to provide an optical recording medium arranged to reduce the spot size and raise the linear recording velocity to raise the recording density and transfer rate and enable satisfactory direct overwriting to be performed without deterioration in the durability against repeated use and stable preservation characteristic of the recorded signal.

[CO20] To achieve the foregoing object, the optical recording medium according to the present invention comprises a recording layer made of at least a phase-change material, wherein a ratio Ac/Aa of absorptance Ac of the recording layer in a state in which the recording layer is crystal with respect to the wavelength of recording/reproducing light and absorptance Aa of the recording layer in a state in which the recording layer is amorphous is 0.9 or higher, and a crystallization enhancing material for enhancing crystallization of the phase-change material is in contact with at least either surface of the recording layer.

[0021] The basic idea of the present invention is to employ both of absorptance control and enhancement of crystal-lization. The two means are combined with each other so that the difference between the physical properties of crystal and those of amorphous is reliably be compensated. Thus, satisfactory direct overwriting can be performed.

[0022] Each of the methods, that is, the absorptance control and the enhancement of crystallization are able to prevent the foregoing problems to an allowable level. Deterioration in the durability against repeated use and the characteristic of stably preserving a recorded signal can be prevented.

[0023] The absorptance control and the enhancement of crystallization are mutually contradictory techniques. If the two techniques are combined with each other, undesirable compensation is expected. Therefore, substantially no attempt of the foregoing combination has been made.

[0024] The inventors of the present invention have found a fact that some countermeasure is required to overcome a problem in that the value of jitters cannot be reduced if the absorptance control is performed under a condition of a high linear velocity. Thus, the inventors have carried out various investigations.

[0025] Among the foregoing investigations, combination of the absorptance control and the enhancement of the crystallization has been investigated. As a result, a fact has been found that their advantages can maximally be realized and problems of each techniques can substantially be prevented if the design is properly performed.

[0026] That is, the absorptance control is performed such that a contrivance of the lamination structure of the film prevents absorption in the amorphous state of the recording layer. As a result, the erasing sensitivity deteriorates in spite of improvement in the durability against repeated reproduction.

[0027] When high speed crystallization is realized by providing the crystallization enhancing layer, the erasing sensitivity can be improved. However, durability against repeated reproduction deteriorates.

[0028] When the foregoing techniques are combined with each other, the recording mark can be erased as required. That is, high-speed overwriting is permitted. When erasing of the recording mark is not required, erasing is inhibited. For example, a crosswriting characteristic can be improved and also the preservation stability can be improved. More-

over, rise in the value of jitters caused from increase in the number of reproducing operations can be prevented.

[0029] The foregoing facts, which have not been expected, are facts which have been found by the inventors of the present invention.

5 Brief Description of Drawings

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- FIG. 1 is a characteristic graph showing change in the duration of the crystallizing temperature occurring due to the difference in the linear velocity and the numerical aperture NA;
 - FIG. 2 is a characteristic graph showing the relationship between the linear velocity and the value of jitters of an optical recording medium having a conventional structure;
- FIG. 3 is a schematic cross sectional view showing an example of a basic structure of an optical disc according to the present invention;
 - FIG. 4 is a schematic view for defining reflectance, transmittance and absorptance of a multilayered film;
- FIG. 5 is a characteristic graph showing dependence of core generating velocity and crystal growing velocity on the temperature;
 - FIG. 6 is a schematic cross sectional view showing an essential portion of an example of an optical disc incorporating a reflecting layer;
 - FIG. 7 is a schematic cross sectional view showing an essential portion of an example of the structure of an optical disc incorporating a light transmission protective layer;
 - FIG. 8 is a schematic cross sectional view showing an essential portion of an example of the structure of an optical disc having multilayered reflecting and recording layers;
 - FIG. 9 is a schematic cross sectional view showing an example of a structure of an optical disc incorporating a dielectric material layer inserted between the layers thereof;
- FIG. 10 is a schematic cross sectional view showing an essential portion of an example of the structure of an optical disc incorporating a light-absorption control layer;
 - FIG. 11 is a schematic cross sectional view showing an essential portion of the structure of an optical disc according to a first embodiment;
 - FIG. 12 is a schematic view showing the difference in the structures among AQCM, QCM and ACM;
 - FIG. 13 is a graph showing waveforms of recording pulses;
- FIG. 14 is a characteristic graph showing change in the values of jitters of the AQCM, QCM and ACM occurring due to change in the number of direct overwriting operations;
 - FIG. 15 is a characteristic graph showing a crosswriting characteristic of the AQCM, QCM and ACM;
- FIG. 16 is a characteristic graph showing the difference in the preservation stability of a recording mark permitted for the AQCM and the QCM;
 - FIG. 17 is a characteristic graph showing change in the values of jitters of the AQCM and the QCM occurring due to a channel clock;
- FIG. 18 is a characteristic graph showing the relationship between the absorptance Aa in an amorphous state and the permitted number of reproducing operations;

FIG. 19 is a characteristic graph showing the number of reproducing operations shown in FIG. 16;

FIG. 20 is a characteristic graph showing dependency of Si-C-H-O on the film thickness;

FIG. 21 is a characteristic graph showing change in the values of jitters realized by performing DOW of SiC and Si-C-H-O with a channel clock of 66 MHz;

FIG. 22 is a characteristic graph showing change in the values of jitters realized by performing DOW of SiC and Si-C-H-O with a channel clock of 76 MHz;

FIG. 23 is a characteristic graph showing the relationship between the number of DOW operations and the values of jitters when Si-N has been formed into a crystallization enhancing layer;

FIG. 24 is a characteristic graph showing the relationship between the number of DOW operations and the values of jitters realized when Al-N is employed to form a crystallization enhancing layer;

FIG. 25 is a characteristic graph showing the relationship between the number of DOW operations and the values of jitters realized when SiO₂ is used to form a crystallization enhancing layer;

FIG. 26 is a schematic cross sectional view showing an essential portion of the structure of an optical disc according to a second embodiment;

FIG. 27 is a schematic cross sectional view showing an essential portion of the structure of an optical disc according to a third embodiment;

FIG. 28 is a characteristic graph showing the relationship between the number of DOW operations and the values of jitters realized by the structure according to the third embodiment;

FIG. 29 is a schematic cross sectional view showing an essential portion of the structure of an optical disc according to a fourth embodiment; and

FIG. 30 is a characteristic graph showing the relationship between the number of DOW operations and values of jitters realized by the structure according to the fourth embodiment.

35 Best Mode for Carrying Out the Invention

[0031] An optical recording medium (an optical disc) according to the present invention will now be described with reference to the drawings.

[0032] FIG. 3 is a diagram showing the basic structure of the optical disc according to the present invention and incorporating a transparent substrate 1 having a thickness of, for example, 0.3 mm or greater. A recording layer 2 is formed on the transparent substrate 1. Moreover, crystallization enhancing layers 3 and 4 are formed in contact with the two surfaces of the recording layer 2.

[0033] The crystallization enhancing layer may be formed on only either surface of the recording layer 2.

[0034] Usually, a phase-change optical recording medium incorporates a reflecting film made of a metal material, such as AI, and having a somewhat large thickness. In the foregoing case, recording/reproducing light is substantially completely reflected by the reflecting layer. A ratio Ac/Aa of the absorptance Ac realized when the recording layer 2 is in a crystal state and the absorptance Aa realized when the recording layer 2 is in an amorphous state is about 0.8.

[0035] The present invention has a structure that the design of films is performed to cause the reflecting layer to be, for example, a transmissive-type reflecting layer which permits somewhat transmission of light. Moreover, the thickness of the foregoing layer, that of the recording layer 2 and that of a dielectric layer to be described later are controlled. Thus, when the recording layer 2 is in a crystal state with respect to the wavelength of the recording/reproducing laser beam (applied to the following description), the ratio Ac/Aa of the absorptance Ac of the recording layer and the absorptance Aa of the recording layer when the recording layer 2 is in the amorphous state is made to be 0.9 or greater. It is preferable that Ac/Aa is 1.2 or greater. It is preferable that the absorptance Aa in the amorphous state is 60 % or lower.

[0036] A plurality of structures are able to realize the Ac/Aa to be 0.9 or greater. One of the methods is to form the reflecting film into a transmission-type reflecting film. Another method is a method which employs a light-absorption control layer. The foregoing structure can be realized without use of the foregoing methods. That is, contrivances of the thickness, the material and the structure (employment of a multilayered structure) of the dielectric material enables the

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foregoing structure to be realized as described later.

[0037] When the ratio Ac/Aa of the absorptance is made to be 0.9 or greater, the crystal portion can relatively easily be heated. Thus, the size of a recording mark which is written on a portion in a crystal state in which no data has been written can be approached to the size of a recording mark which is overwritten on a recording mark in the amorphous state.

[0038] Note that a direction in which the ratio Ac/Aa of the absorptance is raised is a direction in which, for example, the thickness of the reflecting layer is reduced. In the foregoing case, a problem in terms of heat sink arises. Therefore, it is preferable that raising of the ratio Ac/Aa is limited to satisfy a proper range.

[0039] That is, the reflecting layer is usually made of a metal film having a somewhat large thickness, as described above. Therefore, the reflecting layer serves as the heat sink. If the reflecting layer is made of Si or if the thickness of the reflecting layer is reduced, the reflecting layer cannot satisfactorily serve as the heat sink. Therefore, heat cannot easily be radiated. Hence it follows that a heat accumulating structure is realized which deteriorates durability against repeated use.

[0040] Therefore, the ratio Ac/Aa has an allowable range (an upper limit). The allowable range varies depending on the wavelength of recording/reproducing light. In, for example, a present wavelength region (630 nm to 650 nm), it is preferable that the ratio Ac/Aa of the absorptance is 2.0 or lower.

[0041] The absorptance Ac in the crystal state is an absorptance of the recording layer when the recording layer is in the crystal state. Similarly, the absorptance Aa in the amorphous state is an absorptance of the recording layer when the recording layer is in the amorphous state.

[0042] The foregoing values cannot directly be measured when the structure is a multilayered structure. As a result of multiple interference of a multilayered film, the intensity of light which is absorbed into the recording layer must be calculated. Then, the intensity of light must be divided with the intensity of incident light.

[0043] Therefore, the absorptance Ac and the absorptance Aa according to the present invention are defined as follows.

[0044] A multilayered film (an m layer) structured as shown in FIG. 4 is considered. In this case, the reflectance R of the disc, the transmittance T of the disc, the absorptance A of the disc, absorptance A_{l2} of an I_2 layer (for example, the recording layer) and absorptance A_{lm-1} of an I_{m-1} layer (for example, the reflecting layer) are defined as follows.

[0045] Note that an assumption is made that only vertical incident (diagonal incident factors are ignored) is made. Another assumption is made that each interface is smooth and flat (an influence of asperities, such as grooves, is ignored). Moreover, only multiple interference of the multilayered film is factored in. Intensity I of incident light serving as a reference is a value obtained by subtracting intensity Iy of light reflected by the outermost surface from actual intensity Ix of incident light.

Reflectance R of disc:

100 × lr/l (%)

Transmittance T of disc:

100 × It/I (%)

Absorptance A of disc:

100 × (I - Ir - It) (%)

Absorptance A₁₂ of I₂ layer (for example, the recording layer):

100 × I₁₂/I (%)

Absorptance A_{lm-1} of I_{m-1} layer (for example, the recording layer):

 $100 \times I_{lm-1}/I$ (%)

[0046] Note that Ir is the intensity of light reflected by the multilayered film, It is the intensity of light which transmits the multilayered film, I_{l2} is intensity of light which is absorbed by the I_2 layer (for example, the recording layer) and I_{lm-1} is the intensity of light which is absorbed by the I_{m-1} layer (for example, the reflecting layer).

[0047] The absorptance Ac and the absorptance Aa are values obtained by dividing the intensity of light which is absorbed by the recording layer (in the crystal state or the amorphous state) as a result of the multiple interference of the multilayered film with intensity I of light made incident on the multilayered film.

[0048] The foregoing calculation of the multiple interference can be performed by using a characteristic matrix if the wavelength of the laser beam, the thickness of each layer and the complex index of refraction (n - ik) are known.

[0049] A multiplicity of documents about the method of calculating the multiple interference have been published. For example, the following documents are exemplified: "Applied Optics I" (chapter 1-2-2: Intensity of Light) and "Applied Optics II) (Chapter 4-3-2: Multilayered Film) published by Baifu-kan and written by Ikuo Tsuruta and "Thin Optical Film (chapter 3: Multilayered film and Quad-terminal Circuit)" Optical Technique Series II, edited Shiro Fujiwara and published by Kyoritsu.

[0050] As disclosed in the foregoing documents, the relationship can be understood between the electric field and a tangent component of the magnetic field on, for example, a J-th boundary and those on the (J - 1)-th boundary.

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[0051] Energy I of light is expressed by an average |S| of absolute values of pointing vector S. Absorption A_J of a J-th layer is expressed by $A_J = I_{J-1} - I_J$. The absorptance is expressed by $(I_{J-1} - I_J)/I_0$ (where I_0 is incident energy obtained by subtracting the surface reflection component).

[0052] The foregoing calculations are performed by known algorithms. When the algorithms are used, the calculations can easily be performed. In this specification, expression that "only (single layer)" is the foregoing case in which m = 1. [0053] The operations of the crystallization enhancing layers 3 and 4 will now be described.

[0054] In general, crystallization of amorphous proceeds in two stages consisting of generation of crystal core and growth of crystal. An assumption is made that generation of the crystal core randomly occurs in the overall portion in the amorphous phase. As shown in FIG. 5, core generating velocity I is raised as the temperature is raised. The core generating velocity I is raised to a maximum value at a temperature somewhat higher than glass transition point Tg. The core generating velocity I is rapidly lowered at temperatures higher than the glass transition point Tg. On the other hand, the growth velocity U is raised to a maximum value at a temperature close to the melting point Tm.

[0055] In the case of an amorphous mark of an optical disc which is different from the foregoing assumption, the recording material layer 3 has two surfaces. Therefore, generation of the core easily occurs at the interfaces between the recording material layer 3 and layers interposing the recording material layer 3. A fact is known that the chemical or physical properties of the interface exert a great influence on the frequency of generation of the cores.

[0056] Therefore, control of the crystallizing velocity is effectively performed by controlling the states of the interfaces interposing the recording layer 2 to control the generating frequency of the crystal cores.

[0057] Therefore, the present invention is structured such that the crystallization enhancing layers 3 and 4 (only either of the layers is permitted) are formed in contact with the recording layer 2 so that the generating frequency of the crystal cores is controlled. That is, the crystallization enhancing layers 3 and 4 formed in contact with the recording layer 2 enable the generation of the crystal cores to be enhanced so as to raise the crystallizing velocity. Hence it follows that satisfactory overwrite erasing ratio can be realized under the high-speed transfer condition. Thus, satisfactory values of jitters can be obtained.

[0058] Note that the crystallization enhancing layers 3 and 4 may be one layer also serving as the dielectric material layer. For example, a structure may be employed in which the usual dielectric material layer is made of ZnS-SiO₂ (in particular, molar ratio: about 4:1) and a layer for enhancing crystallization is formed on the surface which is in contact with the recording layer 2. As an alternative to this, the dielectric material may be a material which enhances crystallization. That is, the crystallization enhancing layer made of the following material for enhancing crystallization must be in contact with the recording layer 2. Thus, the existence of the dielectric material layer and the type of the material for making the dielectric material layer are not limited. A fact is known that contact of the crystallization enhancing layer with at least either surface of the recording layer 2 raises the crystallizing velocity of the recording layer 2 as compared with the structure having no crystallization enhancing layer. Note that a greater effect of enhancing the crystallization can be obtained when the connection enhancing layers are formed in contact with the two surfaces of the recording layer 2.

[0059] As the material for enhancing the crystallization, known materials (an attenuation coefficient k with respect to the wavelength of the recording/reproducing laser beam is 0.3 or smaller) for use as the dielectric material in the optical recording medium except for a sulphide may be employed. For example, the layer is a layer made of, for example: nitride, oxide, carbide, fluoride, nitroxide, nitrocarbide, oxycarbide of metal of semimetal, such as Al, Si, Ta, Ti, Zr, Nb, Mg, B, Zn, Pb, Ca, La or Ge; or a layer containing the foregoing material as the main component thereof. The layer is exemplified by a layer made of AlN_x (0.5 \le x \le 1) in particular Al₂O_{3-x} (0. \le x \le 1) in particular Al₂O₃; Si₃N_{4-x} (0 \le x \le 1) in particular Si₃N₄; SiO_x (1 \le x \le 2) in particular SiO₂ or SiO; MgO; Y₂O₃, MgAl₂O₄; TiO_x (1 \le x \le 2) in particular TiO₂; BaTiO₃; StTiO₃; Ta₂O_{5-x} (0 \le x \le 1) in particular Ta₂O₅; GeO_x (1 \le x \le 2); SiC; Ge-N; Ge-N-O; Si-N-O; CaF₂; LaF; MgF₂; NaF; or ThF₄; or a layer containing the foregoing material as the main component thereof. The foregoing layers have the function of enhancing the crystallization. Also a layer made of a mixture of the foregoing materials, for example, AlN-SiO₂, has the function of enhancing the crystallization.

[0050] The foregoing materials, however, have not sufficient functions for enhancing crystallization. Therefore, a satisfactory signal characteristics cannot be obtained under severe conditions.

[0061] Therefore, it is preferable that any one of the following materials among the foregoing materials having the function for enhancing crystallization is employed to form the crystallization enhancing layer: Si-C, Si-C-O, Si-C-H, Si-C-H-O, Si-N, Si-N-O, Si-N-H, Si-N-H-O, Si-C-N, Si-C-N-H, Si-C-N-H-O, Si-O, Si-O-H, Al-N or Al-O.

[0052] For example, a material may be employed which is made of Si-C, Si_3N_4 , SiO_2 , AlN, Al_2O_3 or the like or a material may be employed which contains any one of the foregoing materials as the main component thereof and into which an element, such as oxygen, hydrogen or nitrogen, has been introduced.

[0063] When the foregoing material and control of the absorptance (Ac/Aa ≥ 0.9) according to the present invention are combined, a great effect of enhancing crystallization can be obtained. A satisfactory effect of preventing values of jitters can be obtained as described in the embodiment of the present invention. Moreover, improvement in the durability against repeated overwriting operation was confirmed. When the structure according to the embodiment and the mate-

rial which contains SiC as the main component thereof and into which O or H has been introduced are combined, a satisfactory effect was obtained. Also satisfactory crosswriting characteristic and the mark preservation stability were obtained. Moreover, separation at the interface with the recording layer and alloying were prevented. In addition, light absorptivity was reduced. Therefore, the foregoing material was a satisfactory material which is in contact with the recording layer. Therefore, the foregoing materials for enhancing crystallization are preferred materials.

[00:64] Although the thickness of the crystallization enhancing layer is not limited, it is preferable that the thickness is 1 nm or greater to form a uniform film. When the structure according to the embodiment is arranged such that the thickness of the foregoing layer is 2 nm or greater, an effect of reducing values of jitters was obtained.

- [00:65] A method of forming the crystallization enhancing layer is not limited. The following methods are exemplified.
- [0066] SiC can be formed by sputtering, for example, a SiC target by using Ar gas by an RF sputtering method.
- [0067] Si₃N₄ can be formed by sputtering a Si target by using Ar-N₂ gas by a reaction sputtering method.
- [00:68] SiO₂ can be formed by sputtering a SiO₂ target by using Ar gas by an RF sputtering method.
- [00:59] AIN can be formed by sputtering an AI target by using Ar-N2 gas by the reaction sputtering method.
- [0070] Al_2O_3 can be formed by sputtering an Al target by using Ar- O_2 gas by the reaction sputtering method.
- [0071] Si-C-H-O can be formed by sputtering a SiC target by using Ar gas containing H_2O (for example, 300 ppm) by the RF sputtering method.
 - [0072] When the crystallization enhancing layers 3 and 4 are provided, generation of the crystal core is enhanced. Thus, crystallizing speed can be raised. Thus, a great advantage can be realized for direct overwriting.
- [0073] If the crystallizing speed is too high, the preservation stability of the formed recording mark (the amorphous mark) deteriorates. If the crystallizing speed is too low, the effects of the crystallization enhancing layers 3 and 4 cannot be obtained. From the foregoing viewpoints, the crystallizing speed must appropriately be controlled.
- [0074] The optical disc according to the present invention has the foregoing basic structure. The layer structure can arbitrarily be changed.
- [0075] FIG. 6 shows an example in which the reflecting layer 5 is formed. In the foregoing case, the reflecting layer 5 and the recording layer 2 are sequentially formed on the transparent substrate 1.
- [0076] Therefore, recording/reproducing laser beam is applied from a position adjacent to the recording layer 2 so that recording/reproducing is performed. In the foregoing case, it is preferable that a light transmission protective layer is formed on the recording layer 2.
- [0077] An example having the light transmission protective layer is shown in FIG. 7. In this example, the reflecting layer 5, the recording layer 2 and the light transmission protective layer 6 are sequentially formed on the transparent substrate 1. Moreover, the crystallization enhancing layers 3 and 4 are formed in contact with the two surfaces of the recording layer 2.
- [0078] Therefore, the optical disc is irradiated with recording/reproducing light from a position adjacent to the light transmission protective layer 6 so that recording/reproducing with respect to the recording layer 2 is performed.
- [0079] As a matter of course, the structure is not limited to the foregoing structure. A structure may be employed in which the recording layer 2 and the reflecting layer 5 are, in this order, formed on a transparent substrate 1. Thus, recording/reproducing is performed from the position adjacent to the transparent substrate 1. When recording/reproducing is performed from the position adjacent to the light transmission protective layer 4 which has a small thickness, high recording density can be realized because the numerical aperture of the objective lens can be enlarged while a skew margin is being maintained. Therefore, an effect of the application of the present invention can be obtained. Since the recording spot is fined, the temperature of one point on the optical disc is changed in a short time. Therefore, direct overwriting becomes furthermore difficult similar to the rise in the linear velocity.
- [0080] As an alternative to this, the recording layer may be composed of a first recording layer 2A and a second recording layer 2B, as shown in FIG. 8. The reflecting layer may be composed of two layers consisting of a first reflecting layer 5A and a second reflecting layer 5B. When each of the recording layer and the reflecting layer are formed into a multilayered structure having two or more layers, the degree of freedom in optical design is improved. When the reflecting layer is composed of two or more different layers (different in terms of any one of the material, the composition or the complex index of refraction), a dielectric material layer may be inserted into an arbitrary position between the reflecting layer. The number of the dielectric material layers may be determined arbitrarily.
- [0081] For example, as shown in FIG. 9, a dielectric layer 7 (7C, 7B or 7A) may be disposed between the light transmission protective layer 6 and the recording layer 2, between the recording layer 2 and the reflecting layer 5 or between the reflecting layer 5 and the transparent substrate 1. The dielectric material layer is disposed at one or more positions between the foregoing layers. FIG. 9 shows a structure in which the dielectric layers 7 are disposed among all of the layers.
- 55 [0082] When the dielectric layers 7 are disposed while the thickness of each layer is being adjusted, the optical characteristics can be controlled. As a result, the absorptance can easily be controlled.
 - [0083] Note that each dielectric layer 7 may be formed into a multilayered film composed of two or more layers.
 - [0084] As shown in FIG. 10, a light-absorption control layer 8 is formed more adjacent to the position from which

recording/reproducing light is applied as compared with the recording layer 2 to control the absorptance. Also in the foregoing case, the dielectric layer 7 (7D, 7C, 7B or 7A) may be inserted into an arbitrary position including a position between the light-absorption control layer 8 and the light transmission protective layer 6 and a position between the light-absorption control layer 8 and the recording layer 2.

[0085] Also the light-absorption control layer 8 may be composed of two or more different layers (different in terms of any one of the material, the composition or the complex index of refraction). Also in the foregoing case, a dielectric material layer may be inserted into an arbitrary position between light absorptance control layers. Also the number of the dielectric material layers which are inserted may be determined arbitrarily.

[0086] Among the foregoing structures, a preferred structure is exemplified by a structure composed of a substrate/a first dielectric material layer/a reflecting layer/a second dielectric material layer/a first crystallization enhancing material/a recording layer/a second crystallization enhancing material/a third dielectric material layer/a light transmission protective layer.

[0087] In the foregoing case, it is preferable that the thickness of the substrate is 1.0 mm to 1.2 mm, that of the first dielectric material layer is 200 nm or smaller, that of the reflecting layer is 20 nm to 70 nm, that of the second dielectric material layer is 5 nm to 60 nm, that of the first crystallization enhancing material is 2 nm to 20 nm, that of the recording layer is 5 nm to 25 nm, that of the second crystallization enhancing material is 2 nm to 20 nm, that of the third dielectric material layer is 40 nm to 100 nm and that of the light transmission protective layer is 0.05 mm to 0.15 mm.

[0088] When the wavelength of the recording/reproducing laser beam is 630 nm to 650 nm, it is preferable that the thickness of the substrate is 1.0 mm to 1.2 mm, that of the first dielectric material layer is 100 nm to 150 nm or smaller, that of the reflecting layer is 30 nm to 50 nm, that of the second dielectric material layer is 10 nm to 30 nm, that of the first crystallization enhancing material is 2 nm to 10 nm, that of the recording layer is 5 nm to 25 nm, that of the second crystallization enhancing material is 2 nm to 10 nm, that of the third dielectric material layer is 40 nm to 80 nm and that of the light transmission protective layer is 0.05 mm to 0.15 mm.

[0089] As for the material of the transparent substrate 1 of the optical disc having the foregoing structure, a plastic material, such as polycarbonate or acrylic resin, is an advantageous material from a viewpoint of cost reduction. Note that glass may be employed. A manufacturing method may be an injection molding method (an injection method) or a photopolymer method (a 2P method) using a ultraviolet-ray curing resin. Another method may be employed if the method enables a required shape (for example, a disc shape having a thickness of 1.1 mm and a diameter of 120 mm) and satisfactory smoothness and flatness of the surface of the substrate to be realized.

[0090] Although the thickness of the substrate is not limited, it is preferable that thickness is 0.3 mm or larger and 1.3 mm or smaller. If the thickness of the substrate is smaller than 0.3 mm, the strength of the crystal decreases or the disc is easily warped. If the thickness is larger than 1.3 mm, the thickness of the disc is made to be larger than the thickness of 1.2 mm of the CD and the DVD. Thus, there is apprehension that the same disc tray cannot commonly be used when a drive compatible to all of the foregoing mediums is prepared for market.

[0091] When a laser beam is made incident on the light transmission protective layer for performing recording/reproducing, the material of the substrate may be a non-transparent material, such as metal. If the laser beam is made incident on the substrate, a material having substantially no absorptivity with respect to the wavelength of the recording/reproducing laser beam is employed. Note that the material of the substrate may be determined arbitrarily.

[0092] When recording/reproducing light is made incident on the substrate, the order of forming the layers which are interposed between the substrate and the light transmission protective layer is inverted from the forming order employed when recording/reproducing light is made incident on the light transmission protective layer. That is, the technique of the present invention may be applied to the structure of, for example, the CD (the substrate of which has a thickness of 1.2 mm) and the DVD (the substrate of which has a thickness of 0.6 mm).

[0093] A groove track in the form of asperities may be formed on the surface of the substrate on which the multilayered recording film is formed. The grooves are used as guides to enable the laser beam to be moved to an arbitrary position on the disc. The shape of the groove may be any one of a variety of shapes including a spiral shape, a concentric circle, a bit train shape.

[0094] It is preferable that the light transmission protective layer is made of a material having no absorptivity with respect to the wavelength of the recording/reproducing laser beam. Specifically, the material must have a transmittance of 90 % or higher. Moreover, it is preferable that the thickness of the light transmission protective layer is 0.3 mm or smaller. When the thickness is 3 µm to 177 µm and the numerical aperture NA is made to be a large value (for example, 0.85), dense recording not heretofore attained can be realized.

[0095] To prevent adhesion of dust to the surface of the light transmission protective layer and to prevent damage of the same, a protective film made of an organic material or an inorganic material may be formed. Also in the foregoing case, it is preferable that a material is employed which has substantially no absorptivity with respect to the wavelength of the laser beam.

[0096] The optical disc structured such that recording/reproducing is performed from the light transmission protective layer is manufactured by either of the two following methods.

[0097] A first method is a method of laminating a multilayered film on the substrate having the guide groove, followed by finally forming a smooth and flat light transmission protective layer.

[0098] A second method is a method of laminating a multilayered film on the light transmission protective layer having the guide groove, followed by finally forming a smooth and flat substrate.

[0099] The process for forming asperities on the light transmission protective layer or the process for forming the multilayered film is not always an easy process. Therefore, it is preferable that the first method is employed when mass production is factored in.

[0100] When the first method is employed, the light transmission protective layer is formed by bonding a light transmissive sheet (a film), which is made of a plastic material, such as polycarbonate or acrylic resin, having a thickness of 0.3 mm or smaller and which has sufficient optical smoothness by using an adhesive agent (having a thickness of, for example, $5 \, \mu m$ to $15 \, \mu m$) made of ultraviolet curing resin. The bonding process is performed by applying ultraviolet rays. Another method may be employed with which the ultraviolet curing resin is applied to have a required thickness not larger than 0.3 mm by using a spin coater, followed by applying ultraviolet rays.

[0101] If the transmittance is 90 % or higher and the thickness is 0.3 mm or smaller, the structure and the manufacturing method of the light transmission protective layer may be arbitrarily be determined. A groove track in the form of asperities may be provided for the light transmission protective layer. As a method of forming the groove track in the form of asperities in the light transmission protective layer having the thickness of 0.3 mm or smaller, an injection molding method, a photopolymer method (a 2P method) or a method of transferring the asperities by pressing and pressurizing.

[0102] The recording layer 2 of the optical disc having any one of the foregoing structures is made of a phase-change material. That is, the material is a material, the state of which is reversibly changed when the material is irradiated with a laser beam. It is preferable that a material is employed which has the phase which is reversibly changed between the amorphous state and the crystal state. Any one of known materials may be employed which include a chalcogen compound and sole chalcogen.

25 [0103] The materials are exemplified by systems each contains Te, Se, Ge-Sb-Te, Ge-Te, Sb-Te, In-Sb-Te, Ag-In-Sb-Te, Au-In-Sb-Te, Ge-Sb-Te-Pd, Ge-Sb-Te-Se, In-Sb-Se, Bi-Te, Bi-Se, Sb-Se, Sb-Te, Ge-Sb-Te-Bi, Ge-Sb-Te-Co or Ge-Sb-Te-Au. As an alternative to this, a system prepared by introducing an additive in the form of gas of nitrogen or oxygen into the foregoing system. Among the foregoing materials, it is preferable that a material mainly composed of Ge-Sb-Te system is employed. A material may be employed which is prepared by adding an arbitrary element, such as Sb or Pd, to the foregoing material. Another material may be employed which is prepared by adding an element in the form of gas, such as nitrogen or oxygen.

[0104] When the Ge-Sb-Te system is employed and the composition of the material is expressed by a composition formula $GE_xSb_yTe_z$ (where each of x, y and z indicates an atomic ratio of each element), a composition range such that $17 \le x \le 25$, $17 \le y \le 25$ and $45 \le z \le 65$ enables excellent characteristics to be obtained.

[0105] The phase of the foregoing recording layer can reversibly be changed between the amorphous state and the crystal state by adjusting the intensity of the laser beam. The optical change of the reflectance or the like caused from the change in the status is used to perform recording, reproduction, erasing and overwriting. In general, crystallization is temporarily performed (in general, called "initialization") after the film has been formed so as to perform recording/reproducing.

[0106] Note that the recording layer may be constituted by two or more continuous different layers (different in terms of any one of the material, the composition or the complex index of refraction).

[0107] It is preferable that the reflecting layer is made of anyone of a metal element, a semimetal element and a semiconductor having reflectivity with respect to the wavelength of the recording/reproducing laser beam and a thermal conductivity of 0.0004 [J/cm·K·s] to 4.5 [J/cm·K·s], their compound or their mixture. Basically, any one of the known reflecting films for the conventional optical disc may be employed.

[0108] The material is exemplified by a sole element, such as Al, Ag, Au, Ni, Cr, Ti, Pd, Co, Si, Ta, W, Mo or Ge or an alloy mainly composed of the foregoing element. In particular, it is preferable that the Al, Ag, Au, Si or Ge is employed from a viewpoint of practical use. The alloy is exemplified by Al-Ti, Al-Cr, Al-Co, Al-Mg-Si, Ag-Pd-Cu, Ag-Pd-Ti or Si-B. The employed material is employed while the optical characteristics and the thermal characteristics are being factored in. In general, if the thickness of the material is made to be a thickness (for example, 50 nm or larger) with which light transmission is inhibited, the reflectance is raised and heat radiation easily occurs. Since the Al material or the Ag material has a high reflectance (for example, 80 % or higher when $\lambda = 400$ nm) also in a short wavelength region, a preferred result can be realized.

[0109] In this specification, control of the absorptance (Ac/Aa \ge 0.9) is realized by, for example, a structure in which a phase of the laser beam made incident on the light transmission protective layer is caused to transmit through the reflecting layer (the reflecting layer for use in the foregoing case is hereinafter called a "transmissive reflecting layer"). Another structure to be described later may be employed in which the light absorptance control layer is formed at an arbitrary position between the light transmission protective layer and the recording layer which incorporates the crystal-

lization enhancing layer.

[0110] The former structure (the transmissive reflecting layer is employed) encounters limitation of the material and the structure of the reflecting layer. The reflecting layer of the latter structure (the light absorptance control layer is employed) and the other structure (the structure having the multilayered dielectric material layer) is not limited if the foregoing conditions are satisfied (the material made of a metal element, a semimetal element or a semiconductor having the reflectivity with respect to the wavelength of the recording/reproducing laser beam and the thermal conductivity of 0.0004 [J/cm • K • s] to 4.5 [J/cm • K • s], their compound or their mixture).

[0111] The transmissive reflecting layer must satisfy the definition for the reflecting layer (the material made of a metal element, a semimetal element or a semiconductor having the reflectivity with respect to the wavelength of the recording/reproducing laser beam and the thermal conductivity of 0.0004 [J/cm • K • s] to 4.5 [J/cm • K • s], their compound or their mixture). Moreover, it is preferable that the transmissive reflecting layer has a structure that the transmittance of only the reflecting layer (the single layer) with respect to the wavelength of the recording/reproducing laser beam is 10 % or higher and a disc having the foregoing reflecting layer has a transmittance of 1 % or higher. If the transmittance is lower than the foregoing values, a satisfactory effect of controlling the absorptance cannot be obtained. The material is exemplified by a sole element, such as Si, Au or Ge, a compound or a mixture mainly composed of the foregoing element. When Si is employed with respect to a laser beam having a wavelength of 630 nm to 650 nm, the thickness of Si is made to be 20 nm to 70 nm, and, in particular, about 30 nm to about 50 nm to realize satisfactory characteristics. When Au is employed with respect to a laser beam having a wavelength of 640 nm, the thickness of Au is made to be about 5 nm to about 20 nm to realize satisfactory characteristics. Also Al, Ag, Ni, Cr, Ti, Pd, Co, Ta, W or Mo may be employed when the thickness is 10 nm or smaller. If the thickness of Si or Ge is 20 nm or larger, the condition of the transmittance must be satisfied. When a mixture of the foregoing metal, such as Au, or the semimetal and the dielectric material is employed, the transmittance is raised as compared with the material composed of only Au (and having a thickness of about 5 nm to about 20 nm). Therefore, design to enlarge the thickness is permitted. Hence it follows that the degree of freedom to determine the thickness is improved. The mixing ratio may be arbitrarily be adjusted by using the optical constant and the thermal conductivity as the design factors. In the foregoing case, an arbitrary dielectric material (refer to the description about the dielectric material layer which will be performed later). The transmissive reflecting layer is made of a mixture of, for example, a material mainly composed of any one of Au, Al, Ag, Si or Ge and a mixture of ZnS-SiO₂ (having a molar ratio of about 4:1).

[0112] The reflecting layer may be composed of two or more different layers (different in terms of any one of the material, the composition or the complex index of refraction). The foregoing layers may be continued or a dielectric material layer may be interposed between the layers.

[0113] It is preferable that the dielectric layer 7 is made of a material having no absorptivity with respect to the wavelength of the recording/reproducing laser beam. Specifically, it is preferable that a material having an attenuation coefficient k of 0.3 or smaller. The foregoing material is exemplified by a mixture of ZnS-SiO₂ (having a molar ratio of about 4:1). As an alternative to the mixture of ZnS-SiO₂, any one of conventional materials for the optical recording medium may be employed to form the dielectric material layer.

[0114] For example, the layer is a layer made of, for example: nitride, oxide, carbide, fluoride, nitroxide, nitrocarbide, oxycarbide of metal of semimetal, such as Al, Si, Ta, Ti, Zr, Nb, Mg, B, Zn, Pb, Ca, La or Ge; or a layer containing the foregoing material as the main component thereof. Specifically the layer is exemplified by a layer made of AlN_x (0.5 \le x \le 1) in particular AlN; Al₂O_{3-x} (0 \le x \le 1) in particular Al₂O₃; Si₃N_{4-x} (0 \le x \le 1) in particular SiO₂ or SiO; MgO; Y₂O₃, MgAl₂O₄; TiO_x (1 \le x \le 2) in particular TiO₂, BaTiO₃, StTiO₃, Ta₂O_{5-x} (0 \le x \le 1) in particular Ta₂O₅, GeO_x (1 \le x \le 2); SiC; ZnS; PbS; Ge-N; Ge-N-O, Si-N-O; CaF₂; LaF; MgF₂, NaF; and ThF₄. The layer made of any one of the foregoing materials or a layer mainly composed of the same may be employed. As an alternative to this, their mixture, for example, a layer made of AlN-SiO₂, may be employed as the dielectric material layer.

[0115] Note that the dielectric material layer may be constituted by two or more different layers (different in terms of any one of the material, the composition or the complex index of refraction).

[0116] When the structure is employed in which the two or more dielectric material layer are continuously formed, it is preferable that the difference in the absorptance n (assuming that the complex index of refraction with respect to the recording/reproducingy laser beam is n - ik) between the two adjacent dielectric material layers is large from a viewpoint of the optical design (for example, Ac/Aa can easily be raised). In particular, it is preferable that the difference is 0.2 or greater. When a material, such as SiO₂, LiF, MgF₂, CaF₂, NaF, ZnS or TiO₂ is formed adjacent to ZnS-SiO₂ (having a molar ratio of 4:1), the absorptance ratio Ac/Aa can be raised. When the dielectric material layer is constituted by three or more continuous layers, the layers may be constituted by different (different in terms of any one of the material, the composition or the complex index of refraction) layers. As an alternative to this, a structure may be employed in which two or more identical layers are inserted such that ZnS-SiO₂/SiO₂/ZnS-SiO₂. In any case, the number of the layers is not limited if the adjacent dielectric material layer are different from each other (different in terms of any one of the material, the composition or the refraction). Since the optical design freedom is improved as the number of the layers increases, Ac/Aa can easily be raised. The multilayered dielectric material layer may be formed at any one of the illus-

trated positions. When the position is between the light transmission protective layer and the recording layer, a great optical effect can be obtained.

[0117] As described above about the reflecting layer, this specification is attempted to realize the control of the absorptance (Ac/Aa ≥ 0.9) by employing a structure in which the light absorptance control layer is formed at an arbitrary position between the light transmission protective layer and the recording layer which incorporates the crystallization enhancing layer. The light absorptance control layer has the absorptivity with respect to the wavelength of the recording/reproducing laser beam. Moreover, the light absorptance control layer is made of the material, such as a metal element, a semimetal element, a semiconductor, their compound or their mixture which causes the transmittance of only the light absorptance control layer (the single layer) is 3 % or higher and the transmittance of only the light absorptance control layer (the single layer) is 3 % or higher and the transmittance of only the light absorptance control layer (the single layer) is 20 % or higher, the effect to control the absorptance can be improved. If the transmittance of the single layer is 3 % or lower, the recording and erasing sensitivity deteriorates. The light absorptance control layer may be made of a sole element, such as Al, Ag, Au, Ni, Cr, Ti, Pd, Co, Si, Ta, W, Mo, Ge or the like or an alloy mainly composed of the foregoing material. Among the foregoing materials, it is preferable that a compound or a mixture mainly composed of a sole element, such as Au, Al, Ag, Si or Ge is employed from a viewpoint of practical use. When Au, Al or Ag is employed, it is preferable that the thickness is 3 nm to 30 nm from an optical viewpoint.

[0118] When Au or Si is employed with respect to, for example, a red laser beam having a wavelength of 640 nm, the thickness of Au is made to be 3 nm to 30 nm and that of Si is made to be 3 nm to 50 nm.

[0119] Another material, for example, Ni, Cr, Ti, Pd, Co, Ta, W or Mo may be employed if the thickness is made to be, for example, 15 nm or smaller. When a mixture of the foregoing metal, such as Au, or the semimetal and the dielectric material is employed, the transmittance is raised as compared with the material composed of only Au. Therefore, design to enlarge the thickness is permitted. Hence it follows that the degree of freedom to determine the thickness is improved. The mixing ratio may be arbitrarily be adjusted by using the optical constant and the thermal conductivity as the design factors. In the foregoing case, an arbitrary dielectric material (refer to the description about the dielectric material layer which will be performed later).

[0120] The light absorptance control layer may be made of a mixture of, for example, a material mainly composed of any one of Au, Al, Ag, Si or Ge and a mixture of ZnS-SiO₂ (having a molar ratio of about 4:1).

[0121] The linear recording velocity may be made of two or more different (different in terms of any one of the material, the composition or the complex index of refraction) and continuous layers. The foregoing layers may be formed into continuous layers or a dielectric material layer may be interposed between the foregoing layers.

[0122] The wavelength of the recording/reproducing laser beam for use with the optical recording medium according to the present invention may arbitrarily be determined. For example, a red laser beam having a wavelength of 630 nm to 650 nm may be employed to perform recording/reproducing. A violet laser beam having a wavelength of 380 nm to 420 nm may be employed to perform recording/reproducing. As a matter of course, the films must be designed to be adaptable to the wavelength of the employed recording/reproducing laser beam.

[0123] When the present invention is applied to a system using a channel clock of 60 MHz or higher, having a large capacity and exhibiting a high transfer rate, a great effect can be obtained. That is, the system using the channel clock of 60 MHz or higher must employ the technique of the present invention.

[0124] Examples of the present invention will now be described in accordance with results of experiments.

Examples (Example 1)

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[0125] As an example of the present invention, an optical disc structured as shown in FIG. 11 was manufactured. The foregoing optical disc incorporated, on a transparent substrate 11, a first dielectric material layer 12, a transmissive reflecting layer 13, a second dielectric material layer 14, a first crystallization enhancing layer 15, a recording layer 16, a second crystallization enhancing layer 17, a third dielectric material layer 18 and a light transmission protective layer 19 which were laminated. The materials and thicknesses of the foregoing layers were as follows:

transparent substrate 11: polycarbonate substrate (having a thickness of 1.2 mm) provided with groove

transmissive reflecting layer 13: Si (having a thickness of 40 nm)

recording layer 16: Ge₂Sb₂Te₅ (having a thickness of 15 nm)

Light Transmission Protective Layer 19: polycarbonate sheet (having a thickness of 0.1 mm)

Crystallization Enhancing Layers 15 And 17: Si-C-H-O (having a thickness of 4 nm)

First Dielectric Material Layer 12: ZnS-SiO₂ (having a thickness of 150 nm)

Second Dielectric Material Layer 14: ZnS-SiO₂ (having a thickness of 20 nm)

Third Dielectric Material Layer 18: Zns-SiO₂ (having a thickness of 60 nm)

[0126] Si-C-H-O for forming the crystallization enhancing layers 15 and 17 was obtained by using a SiC target and Ar gas containing 300 ppm- H_2O by an RF sputtering method. The composition of the foregoing material was analyzed by a RBS (rutherford backward scattering). Results of the analysis was shown in Table 1 (unit: atom%).

[Table 1]

Analyzed Composition of Si-C-H-O film by RBS Method (atom%)

H C O Si
26.0 30.0 18.0 26.0

[0127] The foregoing optical disc was called AQCM. The AQCM was Example 1.

[0128] To make comparisons, a sample (hereinafter called a "QCM") employed only enhancement of crystallization, a sample (hereinafter called an "ACM") employed only the control of the absorptance and a sample having the conventional structure were manufactured. The structures of the AQCM, QCM and ACM were shows in FIG. 12 such that comparisons were permitted.

Structure Films of QCM

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[0129] polycarbonate substrate (1.2 mm)/Al alloy (100 nm)/ZnS-SiO $_2$ (10 nm)/Si-C-H-O (4 nm)/Ge $_2$ Sb $_2$ Te $_5$ (28 nm)/Si-C-H-O (4 nm)/ZnS-SiO $_2$ (85 nm)/polycarbonate sheet (0.1 mm)

Structure of Films of ACM

[0130] polycarbonate substrate (1.2 mm)/ZnS-SiO $_2$ (150 nm)/Si (40 nm)/ZnS-SiO $_2$ (20 nm)/Ge $_2$ Sb $_2$ Te $_5$ (15 nm)/ZnS-SiO $_2$ (60 nm)/polycarbonate sheet (0.1 mm)

Conventional Structure

[0131] polycarbonate substrate (1.2 mm)/Al alloy (100 nm)/ZnS-SiO₂ (17 nm)/Ge₂Sb₂Te₅ (20 nm)/ZnS-SiO₂ (95 nm)/polycarbonate sheet (0.1 mm)

[0132] Optical characteristics (calculated values) of the foregoing samples with respect to the wavelength of 640 nm were shown in Table 2. The complex index of refraction (n - ik) (n: absorptance and k: attenuation coefficient) (measured values) of the materials for use in the calculations were as follows:

	polycarbonate substrate:	n = 1.58
45	polycarbonate sheet:	n = 1.58
	ZnS-SiO ₂ :	n = 2.13
	Si-C-H-O:	n = 1.9
	Ge ₂ Sb ₂ Te ₅ (in a crystal state):	n = 3.72
		k = 3.52
50	Ge ₂ Sb ₂ Te ₅ (in an amorphous state):	n = 3.70
		k = 1.73
	Si:	n = 3.86
		k = 0.34
	Al alloy:	n = 1.35
5 5		k = 6.30
	Au (employed in another example):	n = 0.19
		k = 3.50

[Table 2]

5		Absorptance of Recording Layer (crystal) : Ac	Absorptance of Recording Layer (amorphous) : Aa	Ac/Aa	Reflectance of Disc (c) : Rc
	AQCM	64.4 %	55.7 %	1.16	14.3 %
	QCM	74.3 %	91.2 %	0.82	23.8 %
10	ACM	60.2 %	50.8 %	1.18	16.0 %
	Conventional Struc- ture (for low speed)	75.2 %	86.7 %	0.87	21.8 %
15		Reflectance of Disc	Transmittance of	Transmittance of Disc	Difference in Phase
		(a) : Ra	Disc (c) : Tc	(a) : Ta	(a-c) : Δφ
	AQCM	3.6 %	15.0 %	28.8 %	-0.22 π (rad.)
20	QCM	2.5 %	0.00003 %	0.0001 %	-0.10 π (rad.)
	ACM	4.5 %	16.8 %	31.5 %	-0.16 π (rad.)
	Conventional Struc- ture (for low speed)	5.0 %	0.00005 %	0.0001 %	-0.26 π (rad.)

[0133] The recording/reproducing characteristics of the foregoing samples were evaluated. The following factors were measured.

Factors Evaluated

[0134]

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- (1) Change in values of jitters with respect to the number of direct overwriting (DOW);
- (2) Measurement of crosswrite (DOW on main track 1000 times, and DOW two adjacent tracks by a multiplicity of times);
 - (3) The relationship between values of jitters and preservation duration when a mark subjected to DOW 1000 times was preserved in 90°C-thermostatic chamber (nitrogen atmosphere);
 - (4) The relationship between number of reproducing operations at which the value of jitters was enlarged when a mark subjected to DOW 1000 times was repeatedly reproduced and reproducing power; and
- (5) The relationship between a channel clock for use in a recording operation and the value of jitters after 1000 times of DOW operation.

[0135] The evaluation of the foregoing factors were performed under the following conditions.

Conditions Under which Evaluations were Performed

[0136]

- Wavelength of Laser Beam: 640 nm (semiconductor laser)
- 55 . NA of Objective Lens (two-group lens): 0.85
 - . Linear Velocity: 9.24 m/s

(1, 7) modulation

Channel Clock: 66 MHz

(User Transfer Rate: about 35 Mbps)

Recording Pulse: comb-shape pulse (see FIG. 13 which illustrate an example of 5T pulse. $n \cdot T$ pulse is composed of $n \cdot 1$ pulses and $Pc \le P1 \le Pp$)

Record Erasing Power: Pp = 6.5 mW

P1 = 2.5 mW Pc = 1 mW

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Reproducing Power: Pr = 0.5 mW

15 . Linear Density: 0.21 μm/bit

Track Pitch: 0.45 µm (land and groove recording)

Definition of land and groove: recess portion with respect to recording/reproducing light was land while projection portion is groove (employed data was that for groove unless otherwise specified)

[0137] Also comparative examples were evaluated similarly except for the recording power and the recording pulse were optimized.

[0138] Results of the evaluation were shown in Table 3 and FIGS. 14 to 17.

[0139] The durability was evaluated such that the value of jitters after 1000 times of DOW operations was 13 % or lower was given \bigcirc and that higher than 13 % was given X.

[0140] The cross write characteristics were evaluated such that 1000 time of DOW operations on the main track were performed, followed by performing DOW of the two adjacent tracks 1000 times. Thus, results each encountered a rise of the value of jitters of the main track of 1 % or lower were given mark \bigcirc those encountered a rise in the same of 2 % or higher were given X and those between the foregoing evaluations were given mark \triangle .

[0141] The preservation stability of the mark was evaluated such that a mark subjected to DOW 1000 times was preserved in a 90°C-thermostatic chamber (nitrogen atmosphere) for 100 hours. Samples encountered rise in the value of jitters was 3 % or lower were given mark () and those encountered a rise of 3 % or higher were given X.

[0142] The repeated reproduction characteristic were evaluated such that a mark subjected to 1000 times of DOW operations were repeatedly reproduced with reproducing power of 0.5 mW. Samples encountered rise in the value of jitters, the rise of which was started after 10000 times of the operations were given mark \bigcirc , those encountered rise in the same which was started after 1000 times or smaller were given mark X and samples encountered intermediate values were given mark \triangle .

[0143] The optimum channel clock were evaluated such that the relationship between the channel clock for use in the recording operation and the value of jitters after 1000 times of DOW operations was measured. The substantially central range of the channel clock range with which the value of jitters was made to be 11 % or lower was employed.

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[Table 3]

5		Value of Jitters after 1000 times of DOW	Durability	Crosswrite characteristic
	AQCM	8.5%	0	0
	QCM	8.0%	0	Δ
10	ACM	9.7%	Х	0
70	Conventional Structure (for low speed)	DOW impossible (60 MHz)	DOW impossible (60 MHz)	△ (tested at 30 MHz)
15		Preservation stability of mark	Repeated reproduction characteristic	Optimum channel clock
	AQCM	0	(When Aa was reduced, characteristic was especially improved.)	66 ~ 90 MHz
20	QCM	X	х	30 ~ 60 MHz
	ACM	0	0	30 ~ 45 MHz
25	Conventional Structure (for low speed)	(tested at 30 MHz)	△ (tested at 30 MHz)	20 ~ 35 MHz

[0144] As can be understood from the foregoing results of the evaluations, the AQCM according to the example had superiority.

[0145] As a result, high speed overwrite higher than the high-speed overwrite characteristic (low jitters and excellent durability) which was the advantage of the QCM was realized. Hence it followed that the speed and linear density were raised

[0146] The disadvantages of the QCM, which were unsatisfactory cross write characteristic and repeated reproduction characteristic, were improved by controlling the absorptance (Ac/Aa \geq 0.9). Thus, a fact was confirmed that rise in the track density was permitted.

[0147] Although the same crystallization enhancing layer as that of the QCM was employed, a satisfactory mark preservation stability was realized. As a result, excellent reliability was maintained. The reason for this is considered that the number of crystal cores of AQCM generated in the amorphous mark during the recording operation is smaller than that of the QCM.

40 Investigation of about Absorptance Aa in Amorphous State

[0148] An influence of the absorptance Aa of the recording layer of the optical disc having a structure similar to that of the example (AQCM) in the amorphous state on the recording/reproducing characteristic was evaluated.

[0149] Samples (i) to (iv) each having a film structure which was similar to that of the AQCM and a thickness changed as shown in Table 4 were manufactured. Note that Table 4 also show calculated absorptance Aa of the samples in the amorphous state.

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[Table 4]

	ZnS-SiO ₂ (nm)	Si (nm)	ZnS-SiO ₂ (nm)	Si-C-H-O
(i)	75	30	12	4
(ii)	105	38	19	4
(iii)	150	40	20	4
(iv)	160	60	20	4
			<u></u>	
	Go Sh Te- (nm)	Si_C-H-O (nm)	7nS-SiO ₂ (nm)	Да
(i)	Ge ₂ Sb ₂ Te ₅ (nm)	Si-C-H-O (nm)	ZnS-SiO ₂ (nm)	Aa 36.9%
(i)	Ge ₂ Sb ₂ Te ₅ (nm)	Si-C-H-O (nm)	62	36.9 %
(i) (ii)		ļ		
	14	4	62	36.9 %

[0150] The samples were used to measure the number of reproducing operations at which the enlargement of the reproducing power and the value of jitters was started. Results were shown in FIG. 18.

[0151] The number of reproducing operations at which the enlargement of the value of jitters was started was determined such that change in the value of jitters depending on the number of reproduction operations was measured as shown in FIG. 19. Then, an inflection point at which the value of jitters were rapidly enlarged with respect to the reproducing power was used to perform the determination.

[0152] As a result, when Aa of the AQCM was reduced such that $Aa \le 60$ %, the repeated reproduction characteristic was improved. Thus, furthermore satisfactory reliability was realized. Since intensification of laser noise and reduction in the quantity of light for performing a servo operation caused from reduction in the laser power were prevented, an effect was obtained in that the degree of freedom in designing the drive was improved.

Investigation of Thickness of Crystallization Enhancing Layer

[0153] A structure similar to that according to the example was subjected to a process in which the thickness of Si-C-H-O which was the material for making the crystallization enhancing layer was changed. Then, the value of jitters realized after overwriting operations were performed 1000 times was measured. Results were shown in FIG. 20.

[0154] As can be understood from FIG. 20, if the thickness of the crystallization enhancing layer was 2 nm or larger, an effect of reducing the value of jitters was obtained. When the thickness was 4 nm, the effect was substantially saturated. As a result, the example had the structure that the thickness of the crystallization enhancing layer was made to be 4 nm.

Investigation about Material of Crystallization Enhancing Layer

[0155] Initially, SiC and Si-C-H-O were selected as the materials of the crystallization enhancing layer so that the recording/reproducing characteristics were evaluated.

[0156] In the evaluation, SiC was $Si_{50}C_{50}$. The composition of the Si-C-H-O was as shown in Table 1.

[0157] The film structure of the manufactured optical disc was similar to that of the AQCM according to the example.

[0158] The foregoing optical discs were subjected to evaluating processes to measure change in value of jitters (see FIG. 21) depending on the number of direct overwriting (DOW) operations at a channel clock of 66 MHz (linear velocity of 9.24 m/second) and change in value of jitters (see FIG. 22) depending on the number of direct overwriting (DOW) operations at a channel clock of 76 MHz (linear velocity of 10.6 m/second). The evaluating conditions were similar to those employed in the foregoing example.

[0159] As can be understood from FIGS. 21 and 22, the difference was enlarged when the channel clock was large. When oxygen and hydrogen were introduced into SiC, recording suitable to high linear speed can be performed.

[0160] Then, the material of the crystallization enhancing layer was changed to Si₃N₄, AlN and SiO₂. Then, similar evaluation was performed. Results were shown in FIGS. 23 to 25.

[0161] In any case, the value of jitters after 1000 times of the DOW operations was reduced by 1 % to 2 %. The reason for this is considered that the crystallizing velocity was raised. Also the durability was improved. Although the value of

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jitters temporarily deteriorates until the DOW operations were performed about 10 times, the deterioration was caused from a fad that the initializing condition was not optimized. Therefore, the foregoing deterioration was irrelevant to the essence of the recording/reproducing characteristic.

[0162] The foregoing effect was obtained when Al₂O₃ was employed.

[0163] When the foregoing results of the experiments were considered, it is preferable that the crystallization enhancing layer was made of a material mainly composed of any one of Si-C, Si-N, Si-C-O, Si-C-H, Si-C-H-O, Si-N-H, Si-N-H-O, Si-C-N-D, Si-C-N-H, Si-C-N-H,

Other Examples

Example 2

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[0164] In this example, an optical disc having the light absorptance control layer was employed.

[0165] That is, in this example, an optical disc having a structure shown in FIG. 26 was manufactured. The optical disc according to this example incorporated a transparent substrate 21 on which a reflecting layer 22, a first dielectric material layer 23, a crystallization enhancing layer 24, a recording layer 25, a crystallization enhancing layer 26, a second dielectric material layer 27, a light absorption control layer 28, a third dielectric material layer 29 and a light transmittance protective layer 30 were laminated. Materials and thicknesses of the foregoing layers were as follows.

transparent substrate 21: polycarbonate substrate (having a thickness of 1.2 mm) provided with grooves

reflecting layer 22: Au (having a thickness of 60 nm)

recording layer 25: Ge₂Sb₂Te₅ (having a thickness of 15 nm)

light transmittance protective layer 30: polycarbonate sheet (having a thickness of 0.1 mm)

crystallization enhancing layers 24 and 26: Si-C-H-O (having a thickness of 4 nm)

light absorption control layer 28: Au (having a thickness of 13 nm)

first dielectric material layer 23: ZnS-SiO₂ (having a thickness of 15 nm)

second dielectric material layer 27: ZnS-SiO₂ (having a thickness of 102 nm)

third dielectric material layer 29: ZnS-SiO₂ (having a thickness of 2 nm)

[0166] The optical characteristics of the manufactured optical discs were shown in Table 5.

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[Table 5]

-	Absorptance of Recording Layer (crystal) : Ac	Absorptance of Recording Layer (amorphous) : Aa	Ac/Aa	Reflectance of Disc (c) : Rc
AQCM	83.2 %	62.6 %	1.33	8.9%
	Reflectance of Disc (a) : Ra	Transmittance of Disc (c)	Transmittance of Disc (a)	Difference in Phase (a-c)
AQCM	28.5 %	0.6 %	1.0 %	-0.17 π (rad.)

[0167] The foregoing optical disc was evaluated similarly to that according to the example (Example 1). The value of jitters after 1000 times of direct overwriting operations was 15 % or lower.

Example 3

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[0168] In this example, an optical disc having a transmissive reflecting film was manufactured.

[0169] That is, in this example, an optical disc having a structure shown in FIG. 27 was manufactured. The optical disc according to this example incorporated a transparent substrate 31 on which a first dielectric layer 32, a transmissive reflective layer 33, a second dielectric material layer 34, a crystallization enhancing layer 35, a recording layer 36, a crystallization enhancing layer 37, a third dielectric material layer 38 and a light transmission protective layer 39 were laminated. Materials and thicknesses of the foregoing layers were as follows.

Transparent substrate 31: polycarbonate substrate (having a thickness of 1.2 mm) provided with a groove

transmissive reflective layer 33: Au (having a thickness of 15 nm)

Recording layer 36: Ge₂Sb₂Te₅ (having a thickness of 14 nm)

Light transmission protective layer 39: polycarbonate sheet (having a thickness of 0.1 mm)

Recording layers 35 and 37: Si-C-H-O (having a thickness of 4 nm)

First dielectric material layer 32: ZnS-SiO₂ (having a thickness of 54 nm)

Second dielectric material layer 34: ZnS-SiO₂ (having a thickness of 10 nm)

Third dielectric material layer 38: ZnS-SiO₂ (having a thickness of 54 nm)

[0170] The optical characteristics of the manufactured optical disc were shown in Table 6.

[Table 6]

	Absorptance of Recording Layer (crystal): Ac	Absorptance of Recording Layer (amorphous) : Aa	Ac/Aa	Reflectance of Disc (c) : Rc
AQCM	70.1 %	59.5 %	1.17	8.2 %
	Reflectance of Disc (a)	Transmittance of Disc (c)	Transmittance of Disc (a)	Difference in Phase (a-c)
AQCM.	0.06 %	20.0 %	37.3 %	0.42 π (rad.)

[0171] The foregoing optical disc was evaluated similarly to the example (Example 1). The value of jitters after 1000 times of direct overwriting operations was 15 % or lower. Change in the value of jitters depending on the number of the direct overwriting operation was shown in FIG. 28.

Example 4

[0172] In this example, an optical disc having two layered structure of the transmission reflecting film was manufactured.

[0173] That is, in this example, an optical disc having a structure shown in FIG. 29 was manufactured. The optical disc according to this example incorporated a transparent substrate 41 on which a first dielectric material layer 42, a first transmission reflecting layer 43, a second transmission reflecting layer 44, a second dielectric material layer 45, a crystallization enhancing layer 46, a recording layer 47, a crystallization enhancing layer 48, a third dielectric material layer 49 and a light transmission protective layer 50 were laminated.

55 [0174] Materials and thicknesses of the foregoing layers were as follows:

Transparent substrate 41: polycarbonate substrate (having a thickness of 1.2 mm) provided with grooves

First transmission reflecting layer 43: Au (having a thickness of 8 nm)

Second transmission reflecting layer 44: Si (having a thickness of 40 nm)

Recording layer 47: Ge₂Sb₂Te₅ (having a thickness of 13 nm)

Light transmission protective layer 40: polycarbonate sheet (having a thickness of 0.1 mm)

Crystallization enhancing layers 46, 48: Si-C-H-O (having a thickness of 4 nm)

First dielectric material layer 42: ZnS-SiO₂ (having a thickness of 52 nm)

Second dielectric material layer 45: ZnS-SiO₂ (having a thickness of 20 nm)

Third dielectric material layer 49: ZnS-SiO₂ (having a thickness of 60 nm)

[0175] Optical characteristics of the manufactured optical disc were shown in Table 7.

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[Table 7]

	Absorptance of Recording Layer (crystal): Ac	Absorptance of Recording Layer (amorphous) : Aa	Ac/Aa	Reflectance of Disc (c) :
AQCM	53.5 %	41.7 %	1.28	14.4 %
	Reflectance of Disc (a)	Transmittance of Disc (c)	Transmittance of Disc (a)	Difference in Phase (a-c)
	: Ra	: Tc	: Ta	: Δφ
AQCM	3.8 %	22.2 %	37.6 %	-0.10 π (rad.)

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[0176] The foregoing optical disc was evaluated similarly to that according to the example (Example 1). Thus, the value of jitters after 1000 times of direct overwriting operations was 15 % or lower. Change in the value of jitters depending on the number of the direct overwriting operation was shown in FIG. 30.

[0177] As can be understood from the foregoing description, according to the present invention, raising of the speed (high transfer rate) and enlargement of the capacity, which are two important factors required to improve the performance of the optical disc can be realized. Moreover, reliability which inhibits raising of the speed can simultaneously be improved.

[0178] That is, according to the present invention, improvement in the performance and satisfactory reliability can simultaneously be realized with an excellent balance. Thus, a technique is provided which is required to form a multi-layered structure (the recording layer) of the disc, furthermore reduce the spot size and permit a phase-change recording medium to be employed when a near-field (proximate field) recording.

45 Claims

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1. An optical recording medium comprising:

a recording layer made of at least a phase-change material, wherein

the ratio Ac/Aa of absorptance Ac of said recording layer which is realized when said recording layer is in a crystal state and absorptance Aa of said recording layer which is realized when said recording layer is in an amorphous state is 0.9 or higher, and

a crystallization enhancing layer for enhancing crystallization of the phase-change material is made contact with at least either surface of said recording layer.

2. An optical recording medium according to claim 1, wherein said recording layer is formed on a substrate.

- 3. An optical recording medium according to claim 1, further comprising a reflecting layer.
- 4. An optical recording medium according to claim 1, further comprising a dielectric material layer.
- 5. An optical recording medium according to claim 1, wherein a recording layer having a crystallization enhancing layer is formed on said substrate, and a light transmission protective layer having a thickness of 0.3 mm or smaller is formed on said recording layer having the crystallization enhancing layer, and

recording/reproducing light is applied from a position adjacent to said light transmission protective layer.

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- An optical recording medium according to claim 5, wherein the thickness of said light transmission protective layer is 3 μm to 177 μm.
- An optical recording medium according to claim 6, wherein said light transmission protective layer is made of polycarbonate or ultraviolet curing resin.
 - An optical recording medium according to claim 5, wherein a reflecting layer is formed on said substrate, and a recording layer incorporating a crystallization enhancing layer is laminated on said reflecting layer.
- 20 9. An optical recording medium according to claim 1, wherein the ratio Ac/Aa of the absorptance Ac realized when said recording layer is in the crystal state and absorptance Aa realized when said recording layer in the amorphous state is 1.2 or larger.
- 10. An optical recording medium according to claim 1, wherein the absorptance Aa realized when said recording layeris in the amorphous state is 60 % or lower.
 - An optical recording medium according to claim 1, wherein said crystallization enhancing material is at least any one of materials selected from a group consisting of Si-C, Si-C-O, Si-C-H, Si-C-H-O, Si-N, Si-N-O, Si-N-H, Si-N-H-O, Si-C-N, Si-C-N-O, Si-C-N-H, Si-C-N-H-O, Si-O, Si-O-H, Al-N and Al-O.

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- 12. An optical recording medium according to claim 1, wherein said recording layer contains a material made of Ge-Sb-Te as a recording material.
- 13. An optical recording medium according to claim 8, wherein said recording layer contains $Ge_xSb_yTe_z$ (where each of x, y and z indicates an atomic ratio of each element and $17 \le x \le 25$, $17 \le y \le 25$ and $45 \le z \le 65$).
- 14. An optical recording medium according to claim 1, wherein said recording layer contains nitrogen and/or oxygen.
- 15. An optical recording medium according to claim 3, wherein said reflecting layer has reflectivity in the wavelength region of a recording/reproducing laser beam, said reflecting layer being made of a material having a thermal conductivity of 0.0004 J/cm K s to 4.5 J/cm K s.
 - 16. An optical recording medium according to claim 3, wherein said reflecting layer is made of at least any one of materials selected from a group consisting of metal, semimetal, a semiconductor, their compound and their mixture.
 - 17. An optical recording medium according to claim 3, wherein said reflecting layer has a transmittance with respect to a recording/reproducing laser beam in a single layer state is 10 % or higher, and

the transmittance of the overall body of said optical recording medium is 1 % or higher.

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- 18. An optical recording medium according to claim 3, wherein said reflecting layer is made of sole element of Al, Ag, Au, Si or Ge, a compound or a mixture mainly composed of any one of the foregoing materials.
- 19. An optical recording medium according to claim 18, wherein the wavelength of the recording/reproducing laser beam is 630 nm to 650 nm, said reflecting layer is made of Si and the thickness d of said reflecting layer satisfies 20 nm < d < 70 nm.</p>
 - 20. An optical recording medium according to claim 3, wherein said having is made of a mixture of at least one material

selected from metal, semimetal, their compound or their mixture and a dielectric material.

- 21. An optical recording medium according to claim 20, wherein said dielectric material is mixture of ZnS-SiO₂.
- 22. An optical recording medium according to claim 21, wherein the content of SiO₂ in the mixture ZnS-SiO₂ is 15 mol% to 35 mol%.
 - 23. An optical recording medium according to claim 3, wherein said reflecting layer is formed by continuously laminating a plurality of layers made of different materials.
 - 24. An optical recording medium according to claim 1, wherein said reflecting layer is formed by continuously laminating a plurality of layers made of different materials.
- 25. An optical recording medium according to claim 5, wherein a dielectric material layer is formed at any one of positions between said light transmission protective layer and said recording layer, between said recording layer and said reflecting layer and between said reflecting layer and said substrate.
 - 26. An optical recording medium according to claim 4, wherein said dielectric material layer is made of a dielectric material having an attenuation coefficient of 0.3 or smaller in a wavelength region of the recording/reproducing laser beam.
 - 27. An optical recording medium according to claim 4, wherein said dielectric material layer is constituted by a plurality of continuous layers, and at least any one of the layers is made of a different material.
- 25 28. An optical recording medium according to claim 27, wherein the difference in the refraction n of adjacent dielectric material layer is 0.2 or larger.
 - 29. An optical recording medium according to claim 28, wherein said plural layers include a layer made of the mixture ZnS-SiO₂ and a layer made of SiO₂ (where 1 ≤ x ≤ 2).
 - 30. An optical recording medium according to claim 25, wherein said recording layer contains Ge-Sb-Te as a recording material, said reflecting layer is made of Si, said crystallization enhancing material contains Si-C, and said dielectric material layer is made of mixture ZnS-SiO₂.
- 35 31. An optical recording medium according to claim 25, wherein a first dielectric material layer, a reflecting layer, a second dielectric material layer, a first crystallization enhancing material, a recording layer, a second crystallization enhancing material, a third dielectric material layer and a light transmission protective layer are sequentially formed on a substrate.
- 40 32. An optical recording medium according to claim 31, wherein the thickness of said substrate is 1.0 mm to 1.2 mm, the thickness of said first dielectric material layer is 200 nm or smaller, the thickness of said reflecting layer is 20 nm to 70 nm, the thickness of said second dielectric material layer is 5 nm to 60 nm, the thickness of said first crystallization enhancing material is 2 nm to 20 nm, the thickness of said recording layer is 5 nm to 25 nm, the thickness of said second crystallization enhancing material is 2 nm to 20 nm, the thickness of said third dielectric material layer is 40 nm to 100 nm and the thickness of said light transmission protective layer is 0.05 mm to 0.15 mm.
 - 33. An optical recording medium according to claim 32, wherein the wavelength of the recording/reproducing laser beam is 630 nm to 650 nm, the thickness of said substrate is 1.0 mm to 1.2 mm, the thickness of said first dielectric material layer is 10 nm to 150 nm or smaller, the thickness of said reflecting layer is 30 nm to 50 nm, the thickness of said second dielectric material layer is 10 nm to 30 nm, the thickness of said first crystallization enhancing material is 2 nm to 10 nm, the thickness of said recording layer is 5 nm to 25 nm, the thickness of said second crystallization enhancing material is 2 nm to 10 nm, the thickness of said third dielectric material layer is 40 nm to 80 nm and the thickness of said light transmission protective layer is 0.05 mm to 0.15 mm.
- 55 34. An optical recording medium according to claim 1, wherein a light absorptance control layer is formed on a surface of said recording layer having said crystallization enhancing layer on which a recording/reproducing laser beam is made incident.

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- 35. An optical recording medium according to claim 5, wherein a light absorptance control layer is formed by said light transmission protective layer and said recording layer having said crystallization enhancing layer.
- 36. An optical recording medium according to claim 34, wherein said light absorptance control layer has an absorptance of 3 % or higher in a single layer state with respect to the recording/reproducing laser beam and a transmittance of 20 % or higher.
- 37. An optical recording medium according to claim 36, wherein said light absorptance control layer is made of at least any one of materials selected from metal, semimetal, their compound and their mixture.
- 38. An optical recording medium according to claim 36, wherein said light absorptance control layer is made of any one of Au, Al and Ag and the thickness d of said light absorptance control layer satisfies 3 nm < d < 30 nm.</p>
- 39. An optical recording medium according to claim 36, wherein said light absorptance control layer is made of a mixture of at least a material selected from metal, semimetal, their compound and theft mixture and a dielectric material.
 - 40. An optical recording medium according to claim 39, wherein said dielectric material is a mixture ZnS-SiO₂.
- 41. An optical recording medium accordingt claim 40, wherein the content of SiO₂ in the mixture ZnS-SiO₂ is 15 mol% to 35 mol%.
 - 42. An optical recording medium according to claim 34, wherein said light absorptance control layer is formed by continuously laminating a plurality of layers made of different materials.
 - 43. An optical recording medium according to claim 1, wherein a channel clock for use in a recording operation is 60 MHz.

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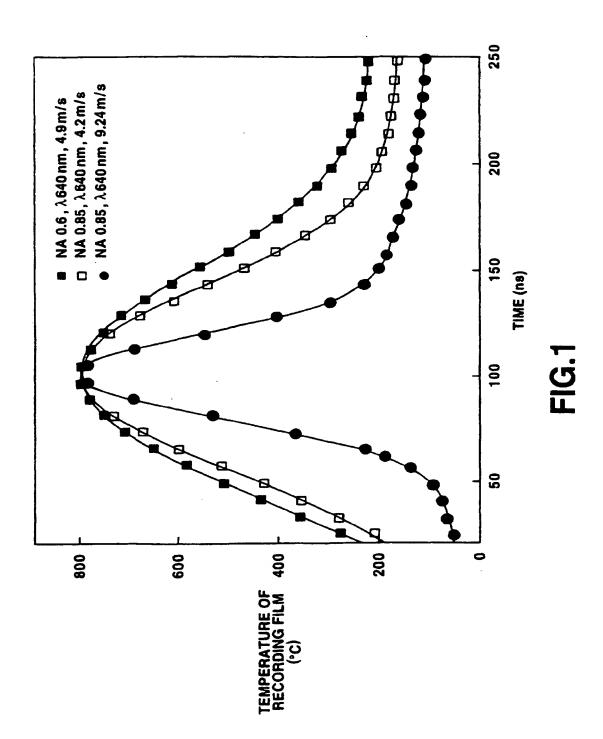
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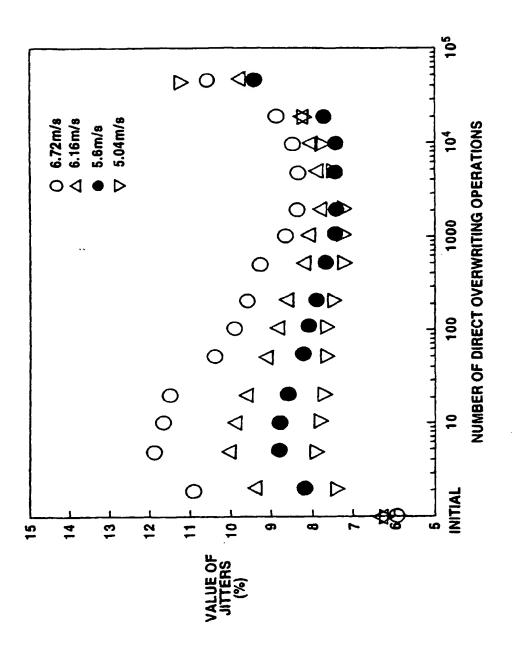


FIG.2

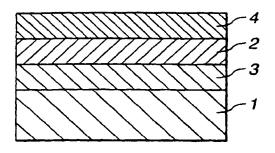
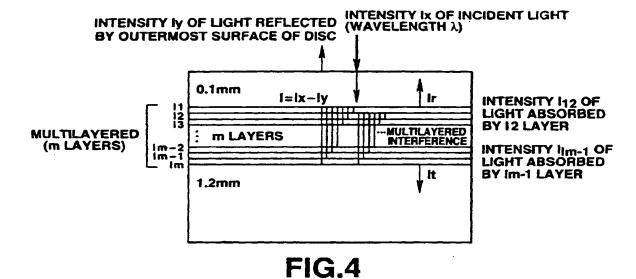
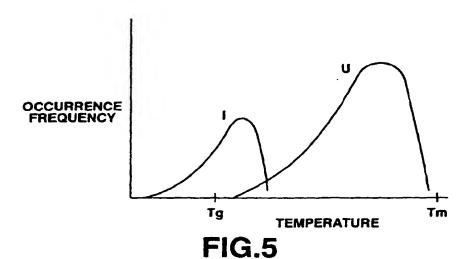


FIG.3





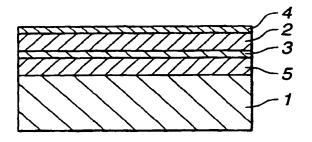


FIG.6

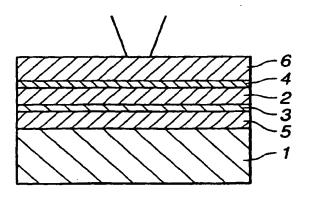


FIG.7

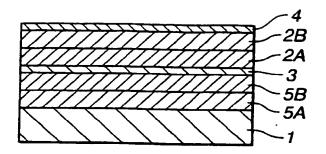


FIG.8

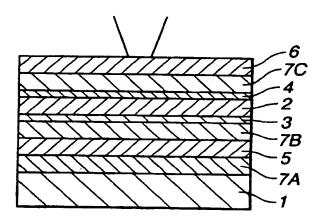


FIG.9

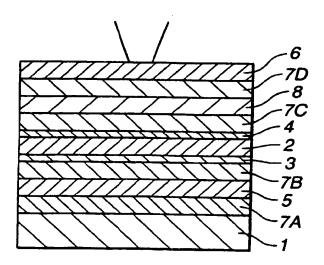


FIG.10

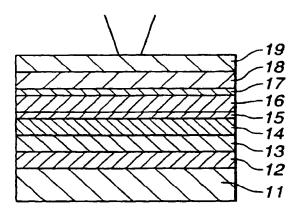


FIG.11

ZnS-SiO ₂
Ge ₂ Sb ₂ Te ₅
ZnS-SiO ₂
Si
ZnS-SiO ₂
ACM

ZnS-SiO ₂	
Si-C-H-O	_
Ge ₂ Sb ₂ Te ₅	
Si-C-H-O	_
ZnS-SiO ₂	_
Al-alloy	_
QCM	

ZnS-SiO2
SI-C-H-O
Ge ₂ Sb ₂ Te ₅
Si-C-H-O
ZnS-SiO ₂
Si
ZnS-SiO ₂
AOCM

FIG.12

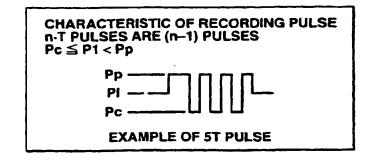


FIG.13

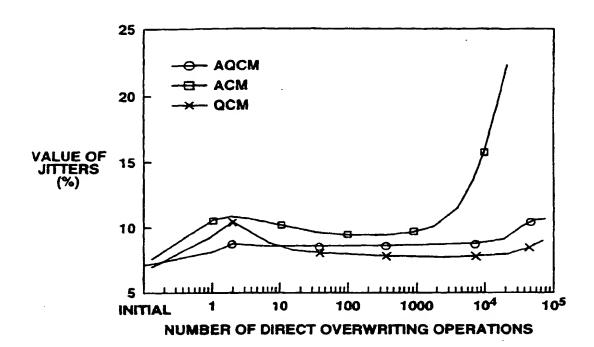


FIG.14

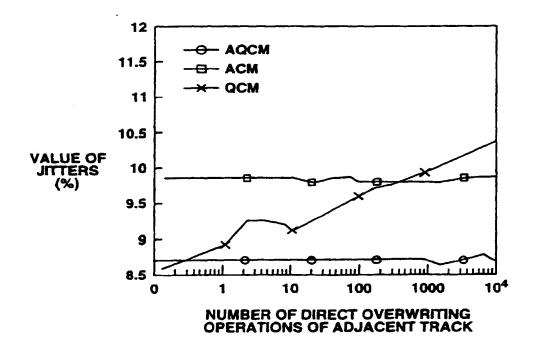
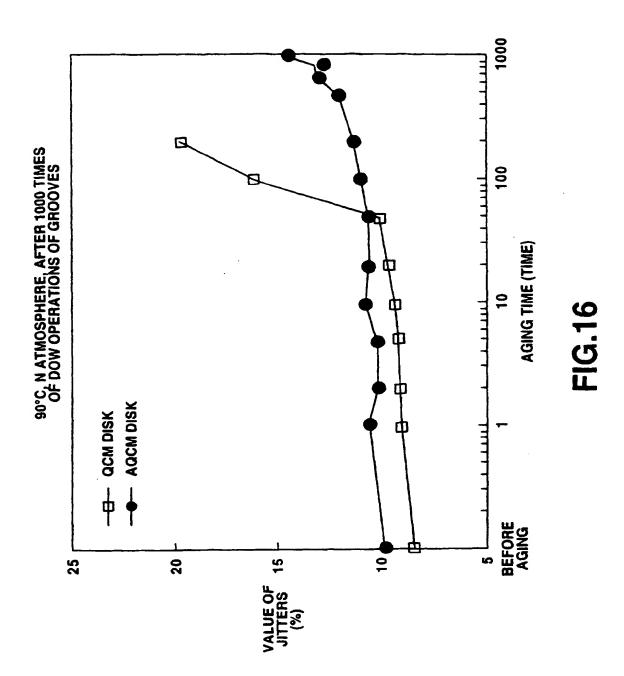


FIG.15



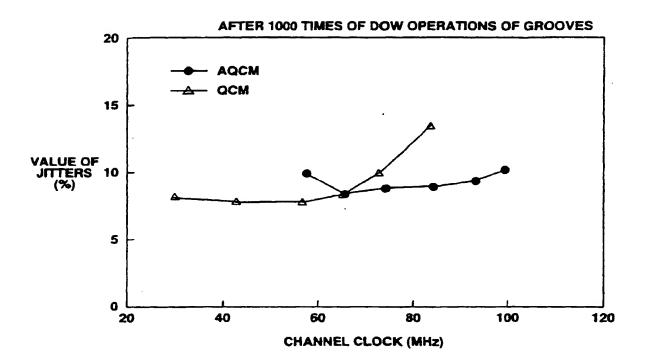


FIG.17

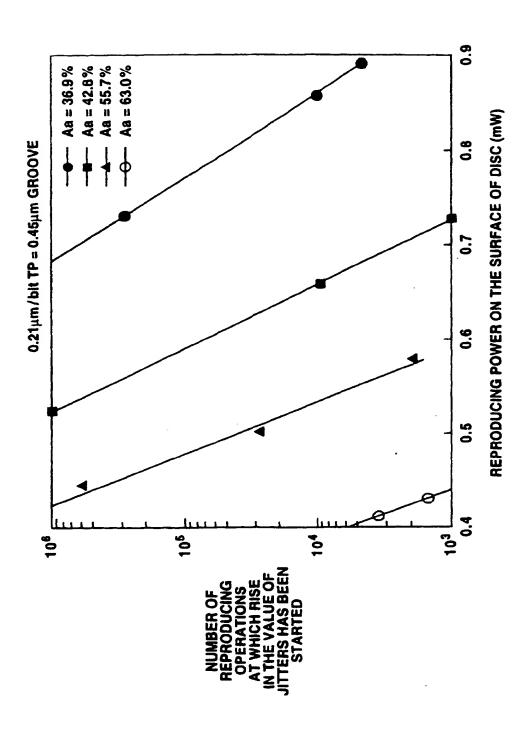


FIG. 18

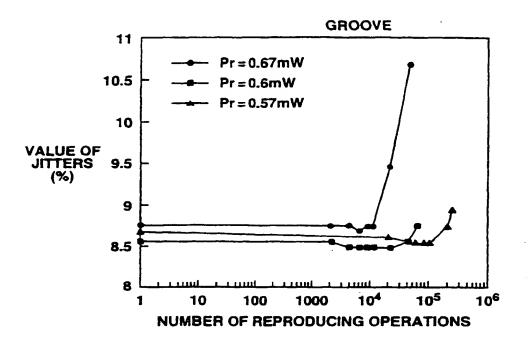


FIG.19

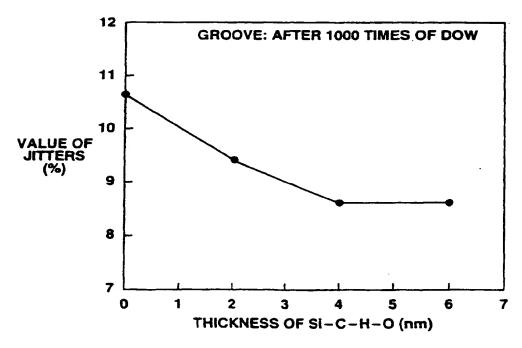


FIG.20

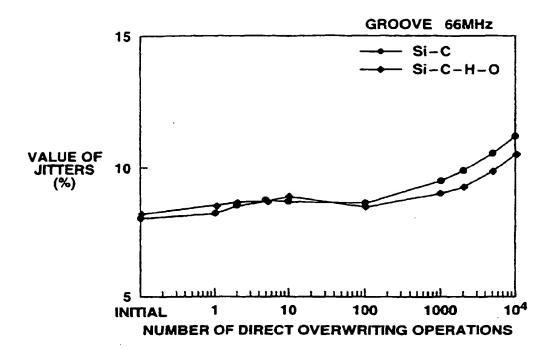


FIG.21

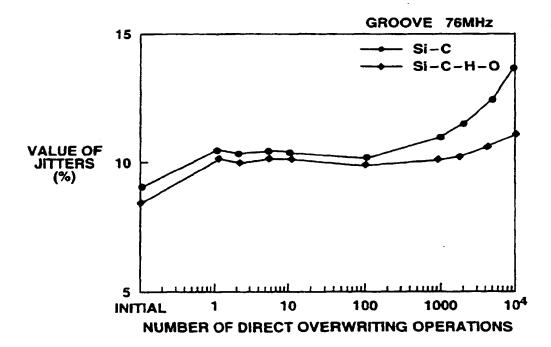


FIG.22

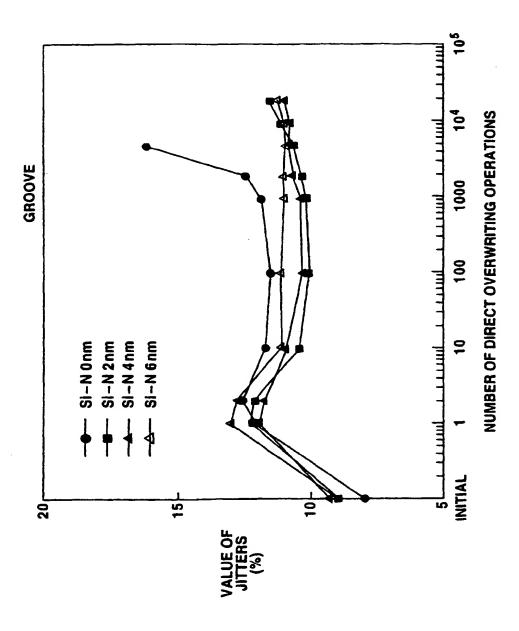


FIG.23

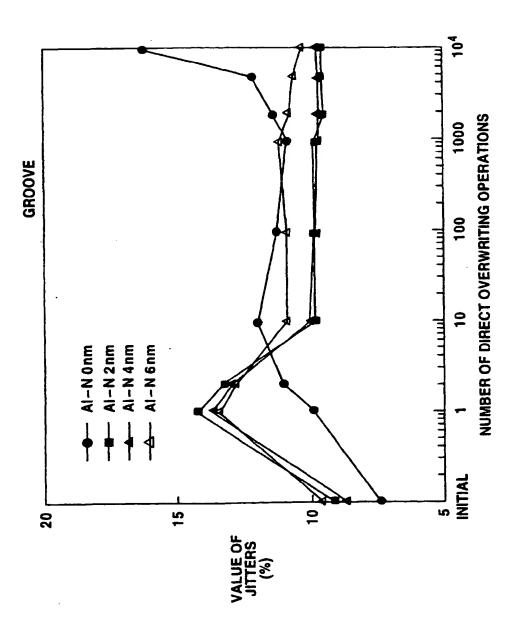


FIG.24

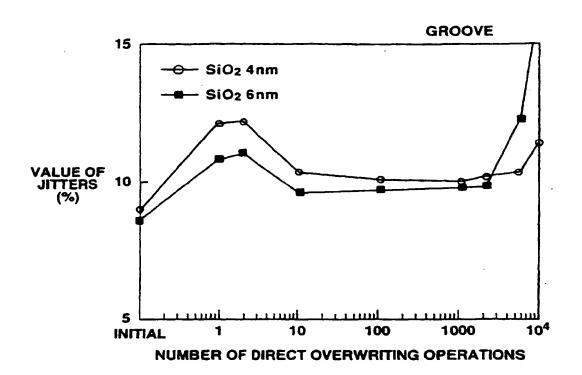


FIG.25

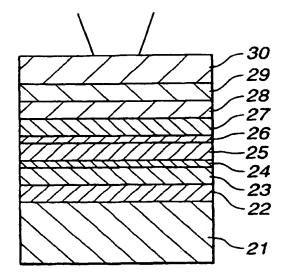


FIG.26

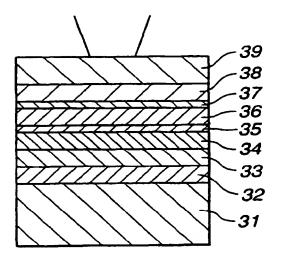
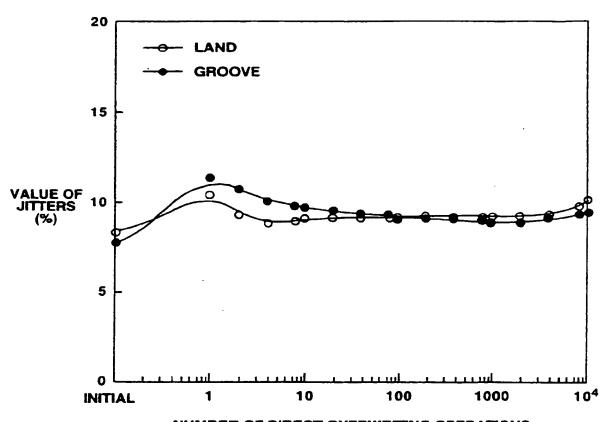


FIG.27



NUMBER OF DIRECT OVERWRITING OPERATIONS

FIG.28

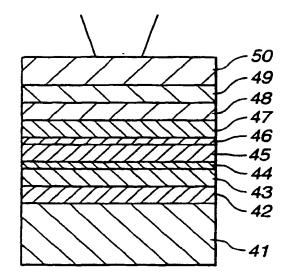


FIG.29

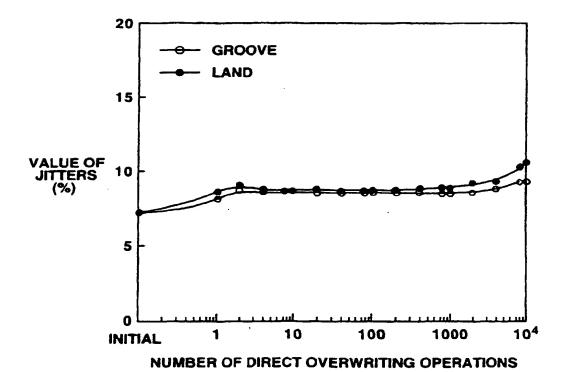


FIG.30

INTERNATIONAL SEARCH REPORT

International application No.
PCT/JP99/00850

A CLASSIFICATION OF SUBJECT MATTER Int.C1 G11B7/24					
According t	According to International Patent Classification (IPC) or to both national classification and IPC				
	6 SEARCHED				
	locumentation searched (classification system followed	by classification symbols)			
Int.	.Cl' G11B7/24				
	tion searched other than minimum documentation to th	e extent that such documents are include	d in the fields searched		
Koka	i Jitsuyo Shinan Koho 1971-1999		1996-1999		
Electronic d	data base consulted during the international search (nar	ne of data base and, where practicable, so	earch terms used)		
C. DOCU	MENTS CONSIDERED TO BE RELEVANT				
Category*	Citation of document, with indication, where ap	propriate, of the relevant passages	Relevant to claim No.		
Y	JP, 8-124218, A (NEC Corp.)		1-43		
	17 May, 1996 (17. 05. 96), Full text (Family: none)				
	JP, 6-195747, A (NEC Corp.)		1-43		
Y	15 July, 1994 (15. 07. 94),	•	1-42		
	Full text (Family: none)				
Y	JP, 9-282713, A (Mitsubishi	Chemical Corp.),	1-43		
	31 October, 1997 (31. 10. 97				
	Full text ; pages 1, 4 (Pami	LIY: none)			
Y	JP, 10-27382, A (Toray Indus		5-8, 13, 25,		
	27 January, 1998 (27. 01. 98 Full text; page 4 (Family:		30-33, 35		
	· - -		, ,		
Y	JP, 9-35332, A (Ricoh Co., 1 7 February, 1997 (07. 02. 97		14		
	Pull text; page 1 (Pamily:				
			L		
<u> </u>	er documents are listed in the continuation of Box C.	See patent family annex.			
	categories of cited documents: and defining the general state of the art which is not	"T" later document published after the inter date and not in conflict with the applica	tion but cited to understand		
	considered to be of particular relevance the principle or theory underlying the invention				
"L" docum	"L" document which may throw doubts on priority chain(s) or which is considered sovel or cannot be considered to involve an investive step cited to establish the publication date of another citation or other when the document is taken alone				
special reason (as specified) "Y" document of particular relevance; the claimed invention cannot be considered to invention on one disclosure, use, exhibition or other considered to invention in inventive step when the document is					
means combined with one or more other such documents, such combination					
*P" document published prior to the international filing date but later than being obvious to a person skilled in the art the priority date claimed "de" document member of the same patent family					
Date of the actual completion of the international search Date of mailing of the international search report					
6 May, 1999 (06. 05. 99) 25 May, 1999 (25. 05. 99)					
Name and mailing address of the ISA/ Authorized officer					
Japanese Patent Office					
Facsimile N	Facsimile No. Telephone No.				

Form PCT/ISA/210 (second sheet) (July 1992)